

## In situ three dimensional monitoring and modelling of small corner cracks in airframe Al alloys

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# Outline

- 1 introduction
- 2 Experimental
  - X-ray tomography experiments
  - Data analysis
- 3 Crack closure model
  - Short presentation of the model
  - Model predictions
- 4 Summary

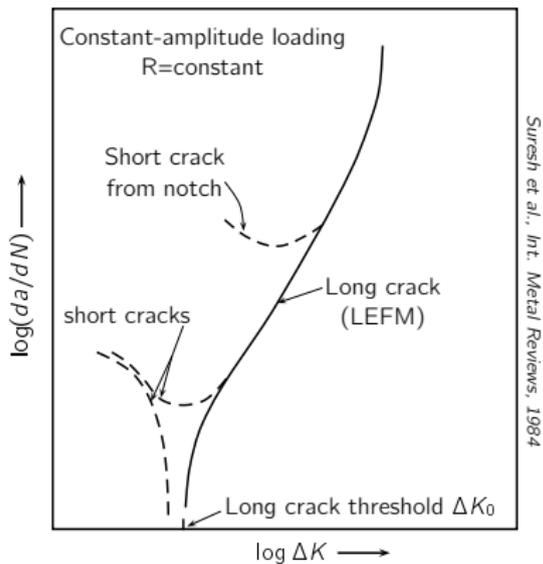


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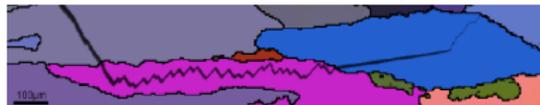


# Short cracks and closure issues



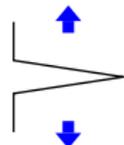
3D crack propagation effects require 3D characterization...

## 1 Crystallographic propagation

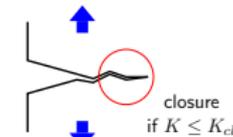


## 2 Crack closure mechanisms

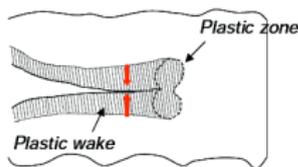
closure free



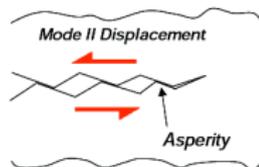
closure affected



$$\text{Driving force } \Delta K_{eff} = K_{max} - K_{cl}$$



'PICC'



'RICC'

# Project aims

## Experimental

Tomographic imaging for detailed insight of small crack evolutions (up to  $\sim 2$  mm)

- Model alloy with ideally flat shape cracks
- Engineering 2027 alloy with highly complex crack shapes

## Propagation model

Apply multi-mechanistic crack closure modelling to small corner cracks

- RICC and PICC
- Constant amplitude, constant  $\Delta K$  and overload regimes
- Plane stress, plane strain and mixed stress state effects

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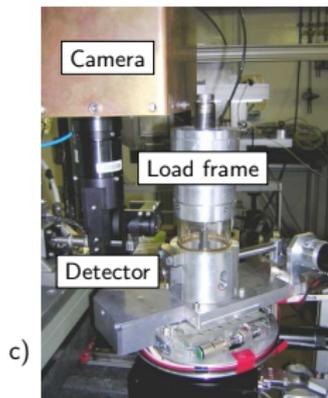
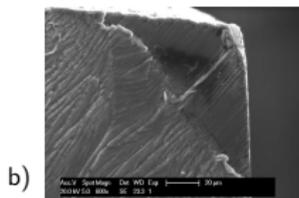
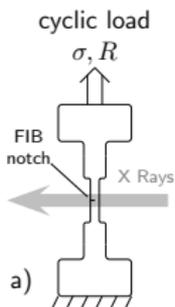


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# in situ X-ray tomography setup



## in situ crack monitoring

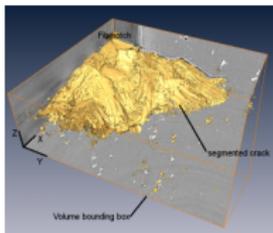
- Two  $\neq$  alloys: 5091 (grain size  $\sim 1 \mu\text{m}$ ) vs. 2027 (grain size  $\sim 100 \mu\text{m}$ )
- crack initiation is controlled via a FIB notch
- in situ* fatigue device
- study both baseline growth ( $\Delta K \sim 3 \text{ MPa}\sqrt{\text{m}}$ ) and post-overload growth

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# Tomographic data analysis

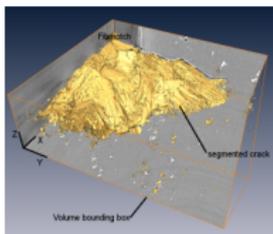


Amount of data is considerable (more than 100 scans) → need automated data treatment as *much as possible* . .

For each tomographic image ( $x,y,z$ ):

- crack front location and local  $da/dN$  values
- local crack opening values
- 3D rendering of crack morphology

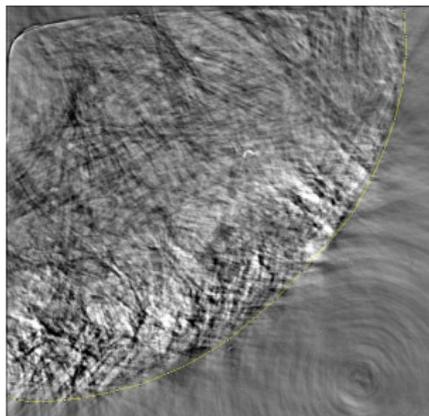
# Tomographic data analysis



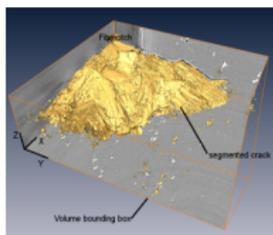
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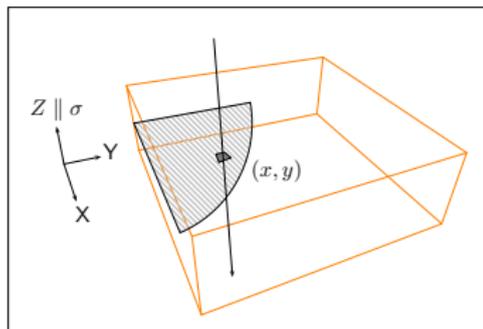
# Tomographic data analysis



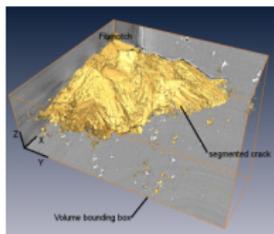
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- **local crack opening values**
- 3D rendering of crack morphology



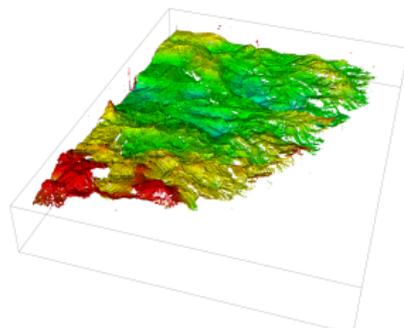
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- local crack opening values
- **3D rendering of crack morphology**







































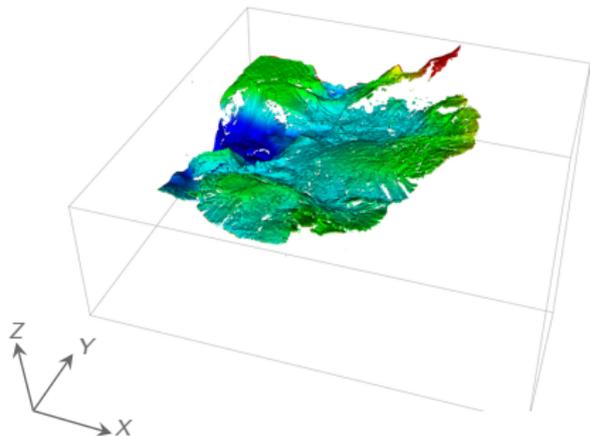


# 3D crack rendering in 2027

N=853k cycles + 100% OL

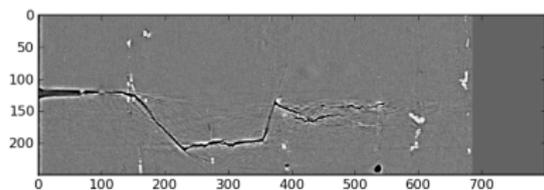
3d rendering with position mapping

box is  $(560 \times 560 \times 175 \mu\text{m})$



slice extracted at  $x = 100$

from raw volume data set



◀ Return

▶ Play

▶ End













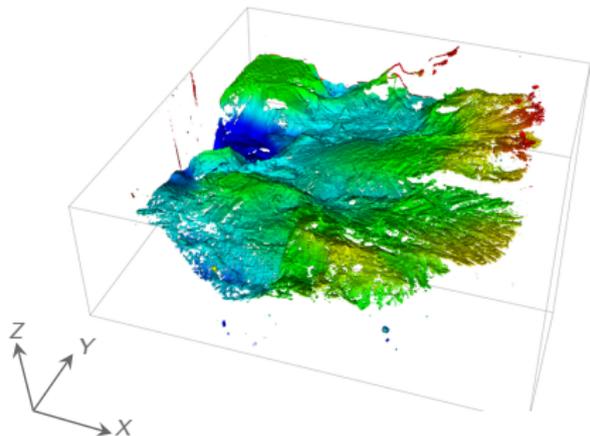


# 3D crack rendering in 2027

N=853k cycles + 100% OL + 59k cycles

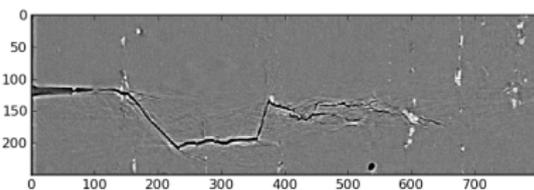
3d rendering with position mapping

box is  $(560 \times 560 \times 175 \mu m)$



slice extracted at  $x = 100$

from raw volume data set

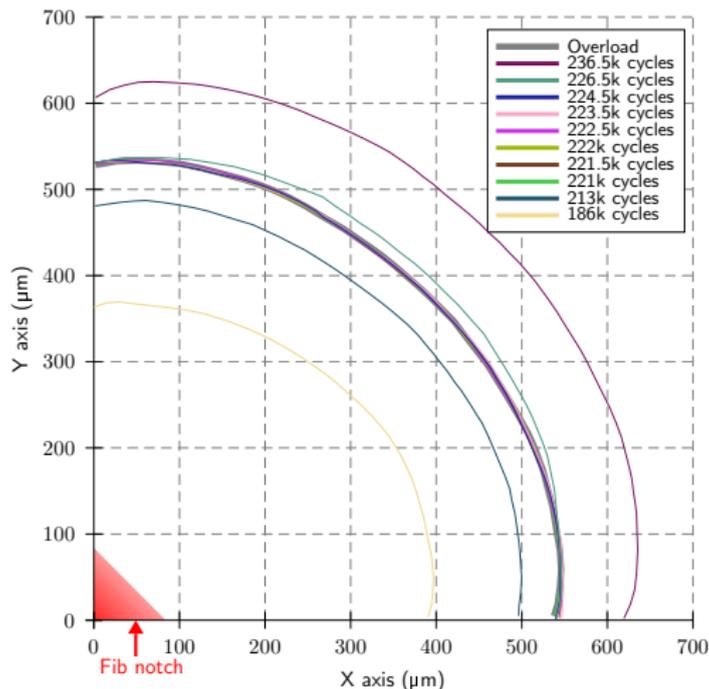


◀ Return

▶ Play

▶ End

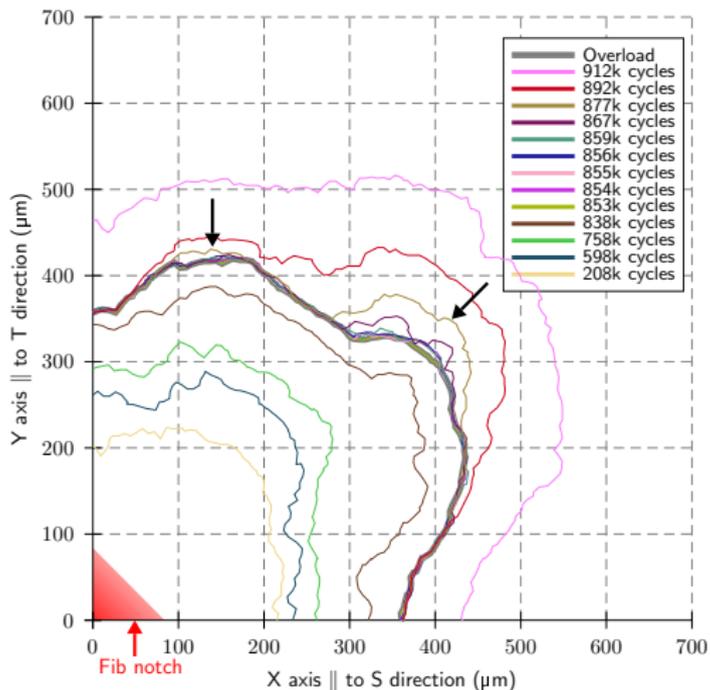
# 5091 vs. 2027 growth behavior



## 5091 crack growth

- Highly planar
- Microstructure independent
- Tunnelling 2-3%
- Overload stopped the crack, growth resumes from the bulk

# 5091 vs. 2027 growth behavior



## 2027 crack growth

- Tortuous crack path
- Strongly microstructure dependent
- Multiple crack branching
- Tunnelling more pronounced
- Overload stopped the crack longer than in 5091, growth resumes from the bulk also

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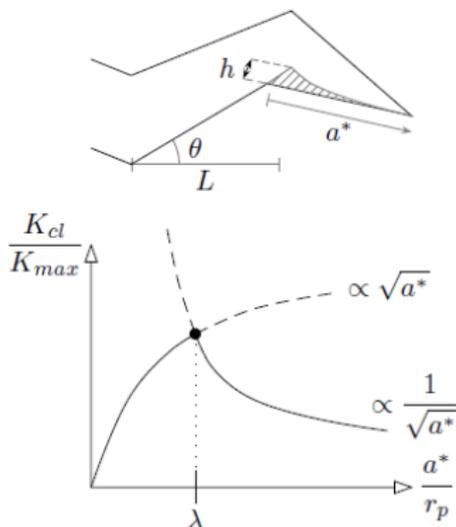
# Model overview

## Extended Southampton multi-mechanistic crack closure model

Apply to both CA growth and post-overload growth with mixed regime plane stress/plane strain description

### $K_{cl}$ determination

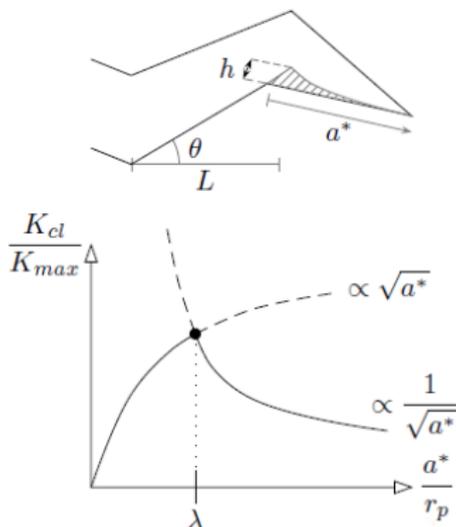
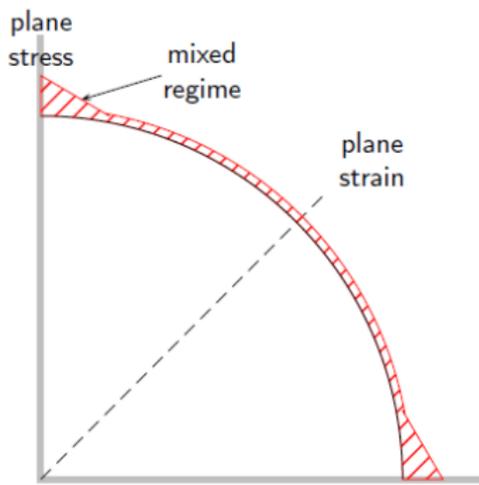
- Plastically deformed material in the crack wake acts as a wedge
- $h$  is directly related to crack opening at the point of overload
- When  $h$  is equal to crack opening, closure occurs
- RICC linked to CTSD, PICC linked to CTOD



# Model overview

## Extended Southampton multi-mechanistic crack closure model

Apply to both CA growth and post-overload growth with mixed regime plane stress/plane strain description



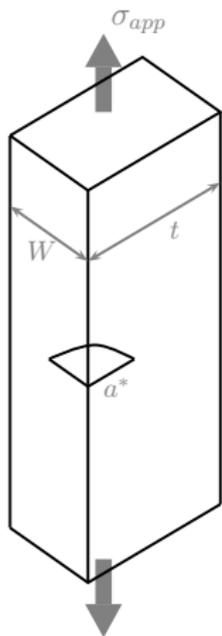
# Model equations

	Plane strain	Plane stress
PICC CA	$\frac{K_{cl}}{K_{max}} = R$	$\frac{K_{cl}}{K_{max}} = 0.47 + 0.53R^2$
PICC OL	$\frac{K_{cl}}{K_{max}} = \beta_I \frac{K_{max} \sqrt{\pi a}}{8\sigma_y(1-\nu^2)\sqrt{l(2a-l)}} \frac{F}{V} (\xi^2 + 2R\xi - 2R - 1)$	$\frac{K_{cl}}{K_{max}} = \beta_I \frac{K_{max} \sqrt{\pi a}}{4\sigma_y \sqrt{l(2a-l)}} \frac{F}{V} (\xi^2 + 2R\xi - 2R - 1)$
RICC CA	$\frac{K_{cl}}{K_{max}} = \beta_{II} \frac{\sqrt{3\pi} K_{max} \left( \sin \frac{\theta}{2} + \sin \frac{3\theta}{2} \right)^2 \sin 2\theta}{32\sigma_y \sqrt{a^*} \left( 3 \cos \frac{\theta}{2} + \cos \frac{3\theta}{2} \right)} \left( \frac{1}{2} + R - \frac{1}{2} R^2 \right)$	
RICC OL	$\frac{K_{cl}}{K_{max}} = \beta_{II} \frac{\sqrt{3\pi} K_{max} \left( \sin \frac{\theta}{2} + \sin \frac{3\theta}{2} \right)^2 \sin 2\theta}{32\sigma_y \sqrt{a^*} \left( 3 \cos \frac{\theta}{2} + \cos \frac{3\theta}{2} \right)} \xi^2 \left( \frac{1}{2} + R - \frac{1}{2} R^2 \right)$	

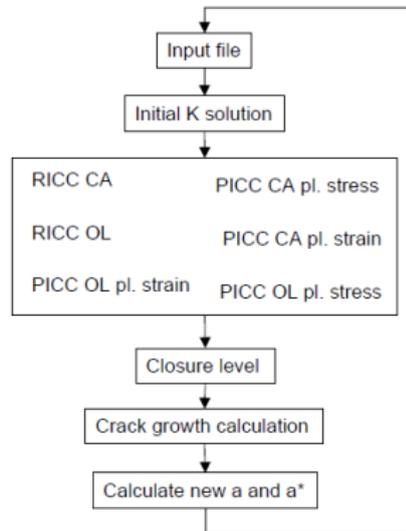
## Mixed state

Closure level is determined at each point of the crack front assuming competition between the two PICC/RICC models.

# Model parameters and flowchart



- Geometrical parameters  
 $a^*$ ,  $t$ ,  $W$
- Materials properties  
 $E$ ,  $\nu$ ,  $\sigma_y$ ,  $L$ ,  $\theta$
- Test parameters  
 $\sigma_{app}$ ,  $R$ -ratio, %OL
- Crack growth parameters  
 $C$ ,  $m$
- Adjustable parameters  
 $\lambda$ ,  $\beta_I$ ,  $\beta_{II}$

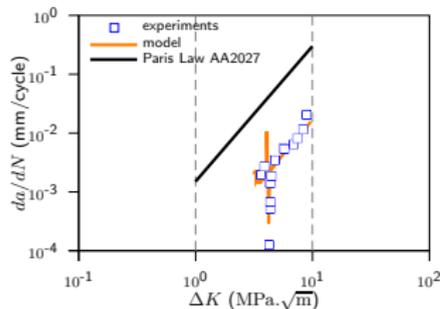
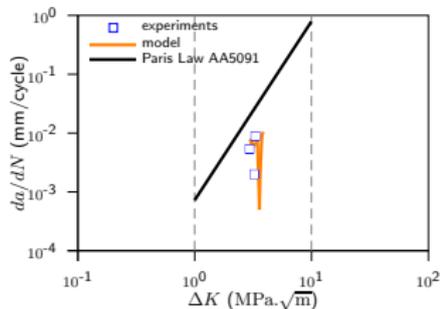
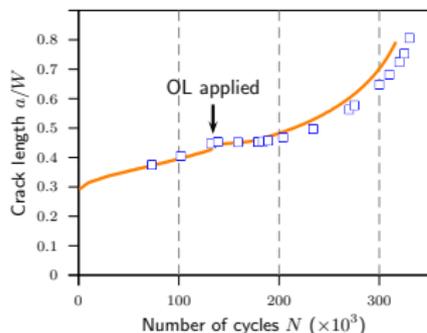
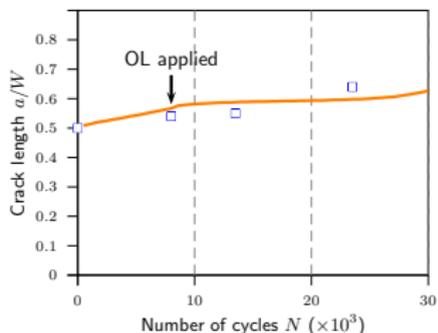


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# Model predictions



The model is able to predict baseline growth and OL transient behavior for both materials within a factor of 2 in terms of number of cycles (equivalent crack length).

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# Summary

## Experimental

- An extensive set of 3D crack propagation data (two materials, baseline growth,  $\neq$  R ratios, OL) has been produced
- Crack growth in the 5091 alloy is mostly microstructure independent
- Crack growth in the 2027 alloy is very dependent on the microstructure and is expected to be predominantly controlled by RICC
- Overloads systematically stop the cracks. After some cycling, the crack growth resume from the bulk (in the middle for the 5091 and from a likely well oriented grain for the 2027)

## Modelling

- Good baseline description for both materials
- 5091 growth is more PICC driven while 2027 is more RICC driven
- Southampton model seems reasonably extendable to small crack description
- Overload retardation can be predicted within a factor of 2 for all experiments with a single set of fitting parameters

# Acknowledgements

THANK YOU

special thanks to:

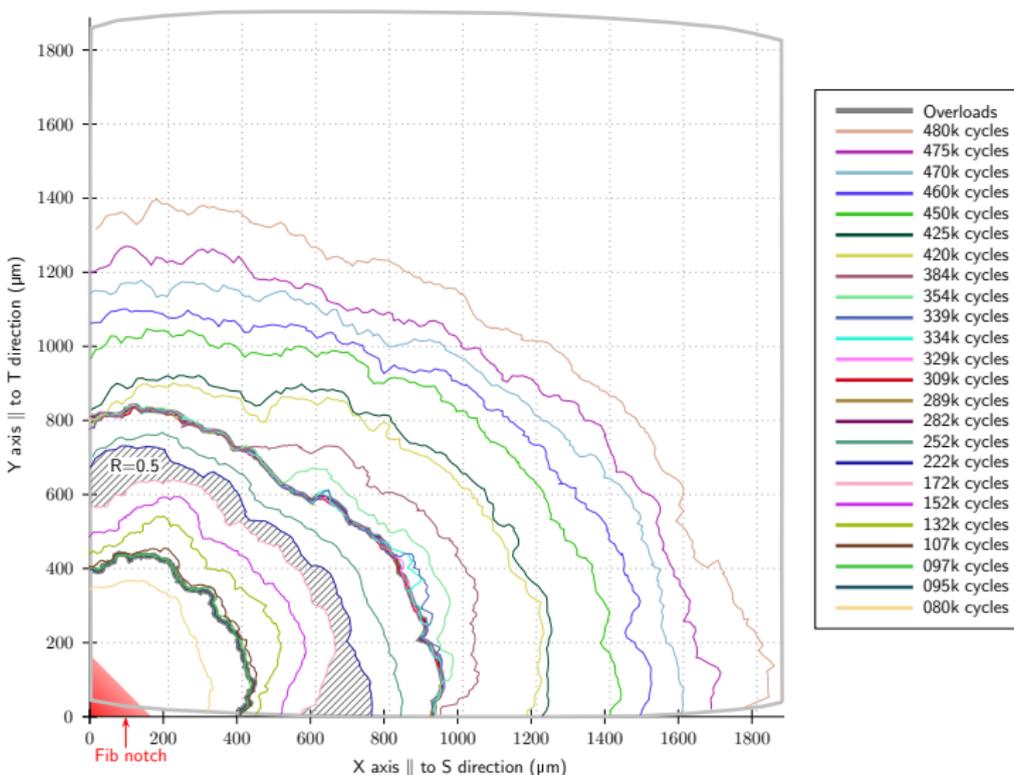
- R. Collins and D. Dai from Airbus UK
- G. Johnson and E. Boller from ID19, ESRF
- Michail Vidiassov for his VTK to U3D exporter

# Outline

## 5 Additional slides

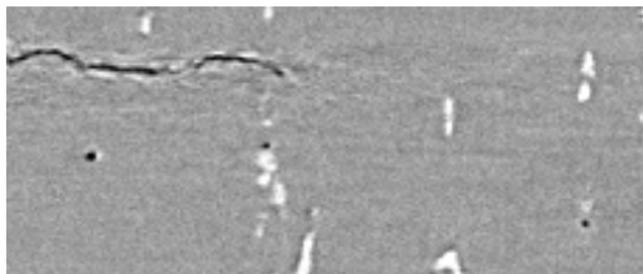
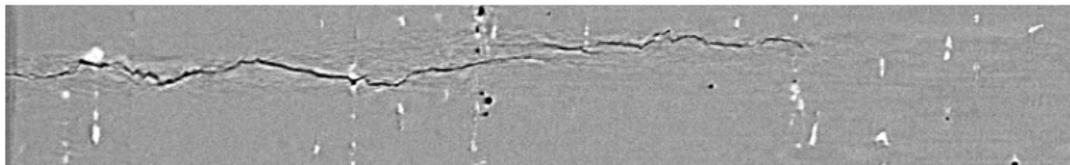


# Crack fronts in $2 \times 2$ 2027 sample



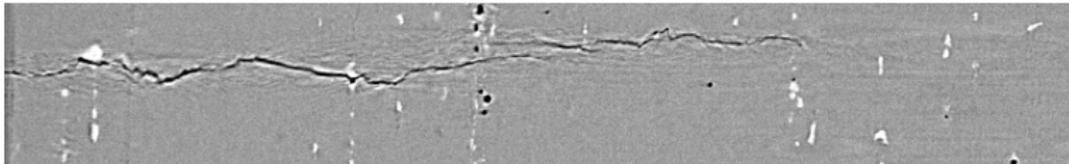
# Overload details

$N = 289k$



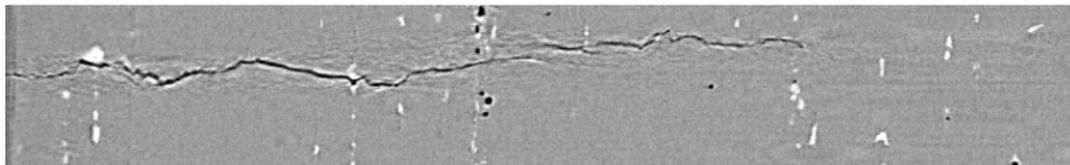
# Overload details

$N = 309k$



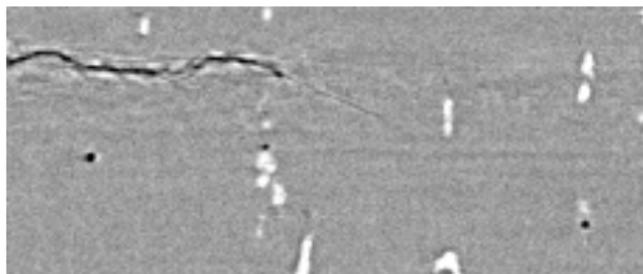
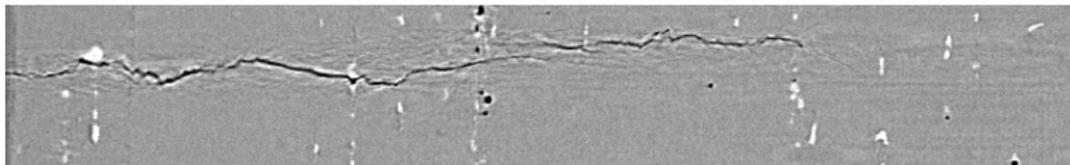
# Overload details

$N = 329k$



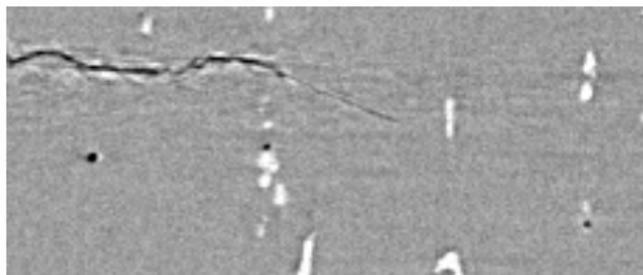
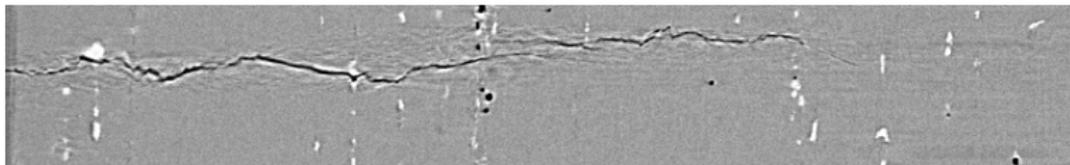
# Overload details

$N = 334k$



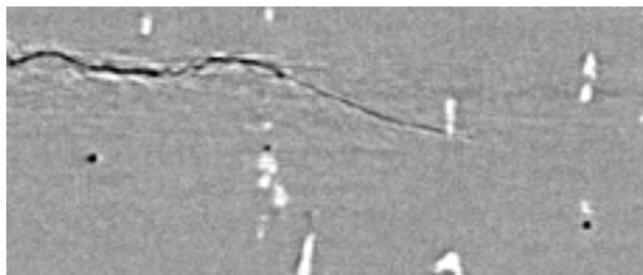
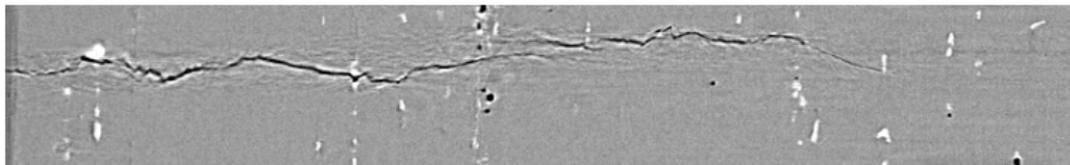
# Overload details

$N = 339k$



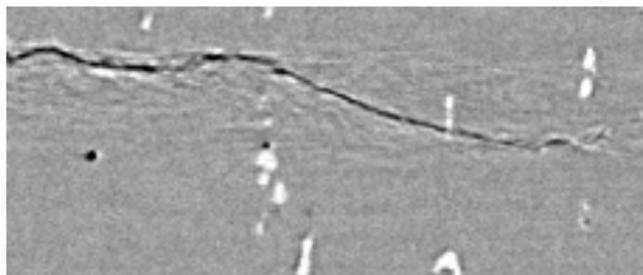
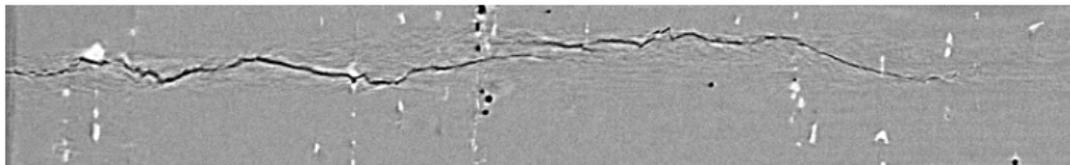
# Overload details

$N = 354k$



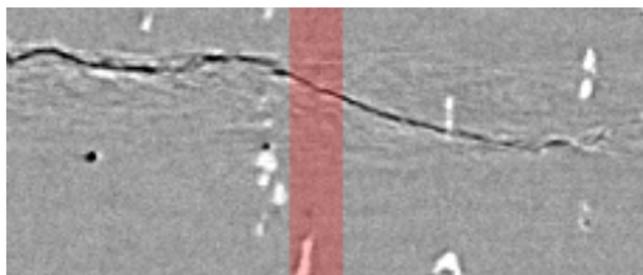
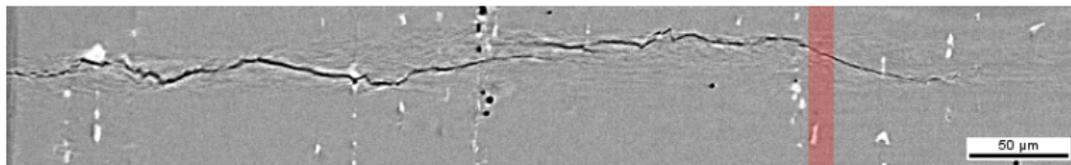
# Overload details

$N = 384k$



# Overload details

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# Model parameters

## Material parameters

∴ 5091 ∴

- $L = 10 \text{ } \mu\text{m}$
- $\theta = 36^\circ$
- Paris law:  
 $m = 3.04 \quad C = 7.2 \cdot 10^{-10}$

∴ 2027 ∴

- $L = 30 \text{ } \mu\text{m}$
- $\theta = 45^\circ$
- Paris law:  $m =$   
 $2.35 \quad C = 13.0 \cdot 10^{-10}$

## Fitting parameters

∴ 5091 ∴

- $\lambda = 0.75/0.42$
- $\beta_I = 1.5/1.3$
- $\beta_{II} = 2.0/2.0$

∴ 2027 ∴

- $\lambda = 0.75/0.42$
- $\beta_I = 1.5/1.3$
- $\beta_{II} = 2.0/3.0$